

Numerical Simulation of the Stress-Strain Behavior of Ni-Mn-Ga Shape Memory Alloys

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joint work with T. Roubíček and P. Šittner



I. Modeling

- Modeling on various length scales
- Elastic energy
- Dissipation
- Evolution equations

II. Numerical Implementation

- Discretization
- Solution method

III. Experimental and Numerical Results

Comparison of laboratory experiments and numerical simulations:

- Evolution of microstructure
- Stress-strain behavior



Modeling of crystalline solids on various length scales:

- **Quantum mechanical level**

 - Electron densities

 - Schrödinger equation and its approximations

- **Atomic level**

 - Atom positions, potential function

 - Newton's equations

- **Continuum mechanical level**

 - Deformation function $y : \Omega \rightarrow \mathbb{R}^3$

 - Potential function, evolution equation

 - Includes „mesoscopical“ models (cf. Young measures, ...)



Interesting points:

- **Upscaling** (Derivation of coarse scale models from fine scale models)
 - Thermodynamic limit (Blanc, Le Bris, P.L. Lions 2002)
 - Direct expansion technique (Kruskal, Zabusky 1964, Collins 1981, Rosenau 1986)
 - Inner expansion technique (A., Griebel 2004)
 - Quasi-continuum method (Tadmor, Ortiz, Phillips 1996)
- **Coupling** of different models within one simulation
 - Bridging Scales Method (W. K. Liu et. al 2003)
 - Heterogeneous Multiscale Method (W. E et. al 2003)
- Analytical Methods:
 - Γ -Limit (Braides et. al 2000)
 - Many other contributions (Friesecke, Theil, Dreyer, ...)



Here:

Modeling of a Ni-Mn-Ga shape memory alloy (SMA) on the **continuum mechanical level**.

(Precisely: Ni-29.1wt.%Mn-21.2wt.%Ga single crystal)

Description of crystal behavior in terms of **energetics**:

- **Elastic energy**
 - Multiwell character: different phases, variants
 - Temperature dependence
 - **Dissipation**
 - Hysteretic behavior
 - Rate independent mechanism
 - **Higher order contributions**
 - Capillarity
 - Viscosity
- not discussed today



Modeling of elastic energy:

- Austenite strain tensor:

$$\varepsilon^0 = \frac{1}{2} (\nabla y^T \nabla y - I)$$

- Martensite strain tensor:

$$\varepsilon^\alpha = \frac{1}{2} (W_\alpha^{-T} \nabla y^T \nabla y W_\alpha^{-1} - I)$$

- Ni-Mn-Ga undergoes cubic to tetragonal transformation.
Wells for austenitic phase and martensitic variants:

$$\begin{aligned} W_0 &= \begin{pmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 1 \end{pmatrix} & W_1 &= \begin{pmatrix} \eta_2 & 0 & 0 \\ 0 & \eta_1 & 0 \\ 0 & 0 & \eta_1 \end{pmatrix} & \eta_1 &= a_M/a_0 = 1.018 \\ & & & \eta_2 &= c_M/a_0 = 0.961 \\ & & & & a_0 &= 5.839\text{\AA} \\ W_2 &= \begin{pmatrix} \eta_1 & 0 & 0 \\ 0 & \eta_2 & 0 \\ 0 & 0 & \eta_1 \end{pmatrix} & W_3 &= \begin{pmatrix} \eta_1 & 0 & 0 \\ 0 & \eta_1 & 0 \\ 0 & 0 & \eta_2 \end{pmatrix} & a_M &= 5.945\text{\AA} \\ & & & & c_M &= 5.610\text{\AA} \end{aligned}$$



- Quadratic form of elastic energy density for each austenite/martensite variant α :

$$\varphi_{\alpha}(\nabla y) = \frac{1}{2} \sum_{i,j,k,l=1}^3 \mathbb{C}_{ijkl}^{\alpha} \varepsilon_{ij}^{\alpha} \varepsilon_{kl}^{\alpha}$$

- Overall elastic energy:

$$\begin{aligned} V_{\text{el}}(y, \theta) &= \int_{\Omega} \varphi(\nabla y(x), \theta) \, dx \\ &= \int_{\Omega} \min_{\alpha=0,\dots,3} (\varphi_{\alpha}(\nabla y(x)) + \psi_{\alpha}(\theta)) \, dx \end{aligned}$$

- Temperature-dependent offset:

$$\begin{aligned} \psi_{1,2,3}(\theta) &= C(\theta - \theta_{\text{eq}}) & C &= \text{Clausius-Clapeyron slope} \\ \psi_0(\theta) &= 0 & \theta_{\text{eq}} &= \text{equilibrium temperature} \end{aligned}$$

- Elastic stress tensor: $\sigma_{\text{el}} = \frac{\partial}{\partial \nabla y} \varphi(\nabla y, \theta)$



Modeling of dissipation:

- Observation:
 - SMAs dissipate a certain amount of energy during each phase transformation.
 - This dissipation is (mostly) **rate-independent**.→ Capture this behavior within our model.

- Introduce **phase indicator functions**

$$\lambda_\alpha : \mathbb{R}^{3 \times 3} \rightarrow \mathbb{R}$$

for each variant $\alpha=0,1,2,3$, which fulfill

- $\lambda_\alpha=1$ nearby of well W_α
- $\lambda_\alpha=0$ far away from well W_α
- smoothly interpolated



- Introduce **dissipation potential**:

$$\xi : \mathbb{R}^4 \rightarrow \mathbb{R} \quad \xi(z) = \sum_{\alpha=0}^3 \mathcal{E}_\alpha |z_\alpha|$$

- Dissipation rate:

$$\xi \left(\frac{\partial}{\partial t} \lambda(\nabla y) \right)$$

constants describing the amount of dissipation

- Dissipated energy over time interval $[t_1, t_2]$:

$$\xi \left(\text{Var}_{[t_1, t_2]} \lambda(\nabla y) \right)$$

(Note: total variation is a rate-independent quantity!)

- Associated **quasiplastic stress tensor**:

$$\sigma_{\text{pl}} \in \partial \xi \left(\frac{\partial}{\partial t} \lambda(\nabla y) \right) \cdot \lambda'(\nabla y)$$



- Putting it together: **Evolution equation**

$$\rho \frac{\partial^2 y}{\partial t^2} = \operatorname{div} \left(\sigma_{\text{el}}(\nabla y) + \sigma_{\text{pl}}\left(\nabla y, \frac{\partial}{\partial t} \nabla y\right) \right)$$

ρ = mass density

- Transformation process in SMA experiments here is very slow
→ Mass density ρ can be neglected.

$$0 = \operatorname{div} \left(\sigma_{\text{el}}(\nabla y) + \sigma_{\text{pl}}\left(\nabla y, \frac{\partial}{\partial t} \nabla y\right) \right)$$



Initial conditions:

- Prescribe deformation and velocity at $t=0$

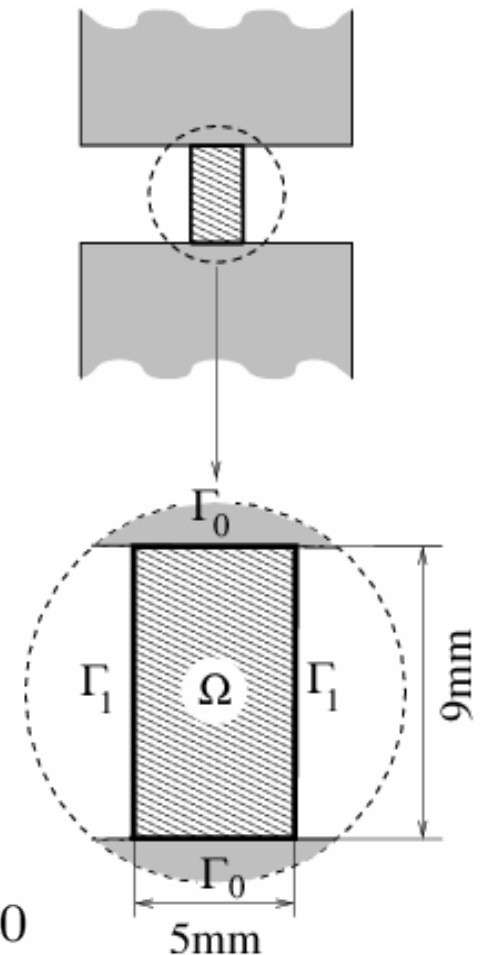
Boundary conditions:

- Time-dependent **Dirichlet** boundary conditions at fixed boundary part Γ_0 :

$$y(x, t) = y_0(x, t), \quad x \in \Gamma_0 \subset \partial\Omega$$

- Homogeneous **Neumann** boundary conditions at free boundary part Γ_1 :

$$\nabla y(x, t) \cdot \nu = 0, \quad x \in \Gamma_1 = \partial\Omega \setminus \Gamma_0$$



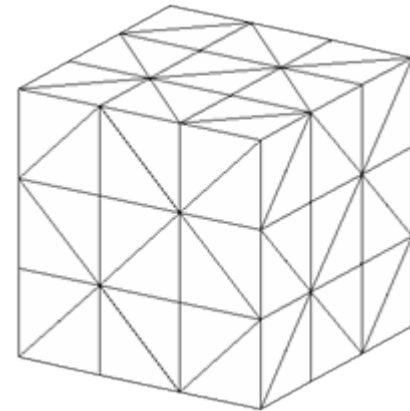


Part II: Numerical Implementation

Goal: Solve evolution equation numerically.

Discretization in space:

- Decomposition of domain Ω into tetrahedra
- **Finite Element** method with P1 Lagrange ansatz functions:
 - piecewise linear on each tetrahedron
 - continuous on whole domain Ω





Discretization in time:

- Subdivide time interval into time slices:

$$0 = t_0 < t_1 < t_2 < \dots < t_{J-1} < t_J$$

- **Finite Difference** method

Solution procedure:

At each time step t_j find $y^{(j)}$ which **minimizes the energy functional**

$$V(y^{(j)}) = V_{\text{el}}(y^{(j)}) + \xi \left(\lambda(\nabla y^{(j)}) - \lambda(\nabla y^{(j-1)}) \right)$$

Theorem: Each (local) minimizer is a solution of the discretized evolution equation.



Minimization algorithm: **Gradient method.**

At each time step j :

- Line search: find minimum along line

$$y^{(j-1)} - s^{(j)} \nabla V(y^{(j-1)}), \quad s^{(j)} \in \mathbb{R}$$

- Determination of step size $s^{(j)}$ by **modified Armijo method:**

$$s^{(j)} = s^{(j-1)} \max\{2^i : i \in \mathbb{Z} \text{ and } \forall k \in [\min(0, i), i] \cap \mathbb{Z} :$$

$$V\left(y^{(j-1)} - 2^k s^{(j-1)} \nabla V(y^{(j-1)})\right)$$

$$\leq V(y^{(j-1)}) - \beta 2^k s^{(j-1)} \|\nabla V(y^{(j-1)})\|^2$$

- Repeat this several times



Gradient method is a **local** minimization technique.

Improve minimization algorithm to find better minimum:

Employ **simulated annealing** technique:

At each time step j :

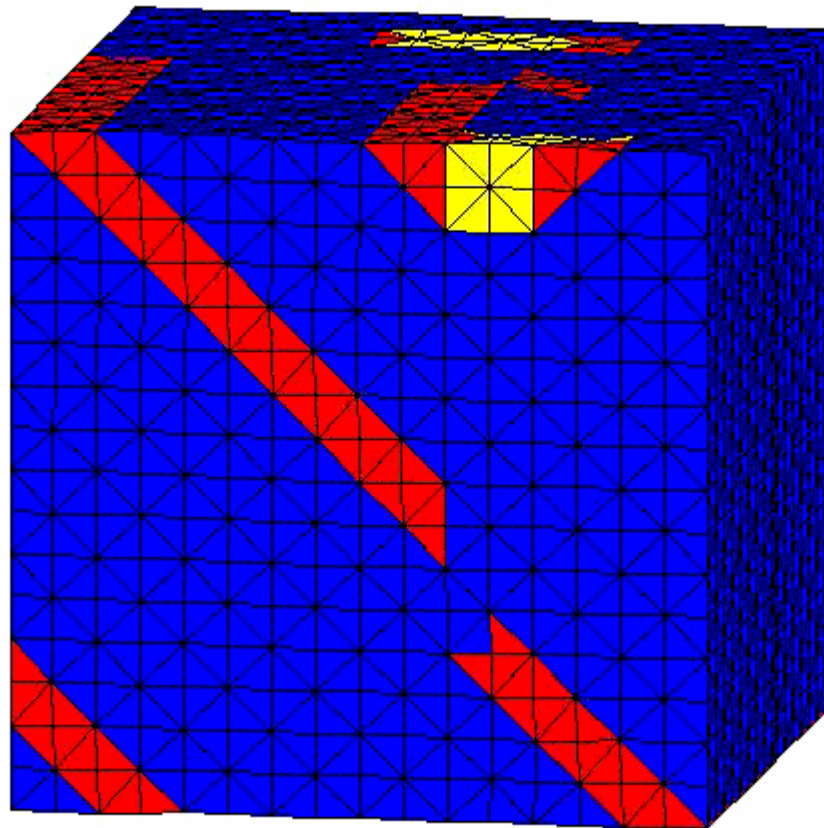
- generate random perturbation y^* from $y^{(j-1)}$
- if $V(y^*) < V(y^{(j-1)})$: always accept
- otherwise: accept with probability

$$\exp\left(\frac{V(y^{(j-1)}) - V(y^*)}{k}\right)$$

- Repeat this several times
- Local minimization with gradient method



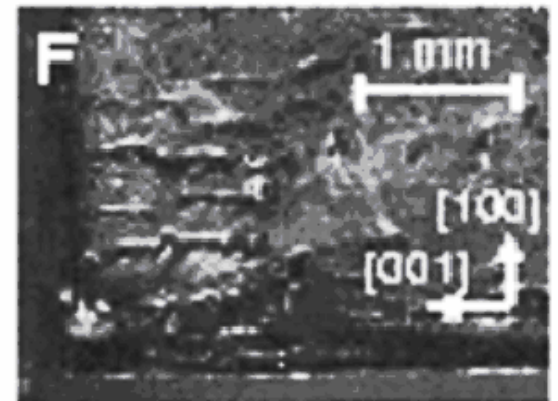
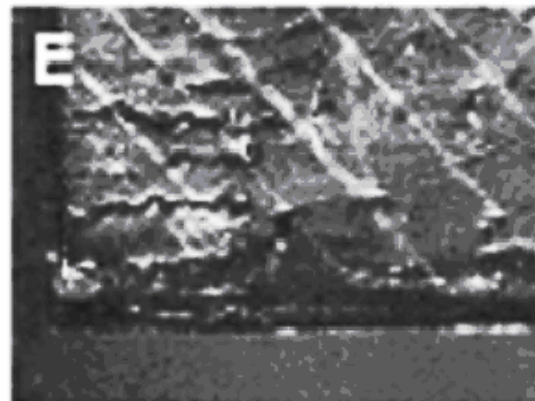
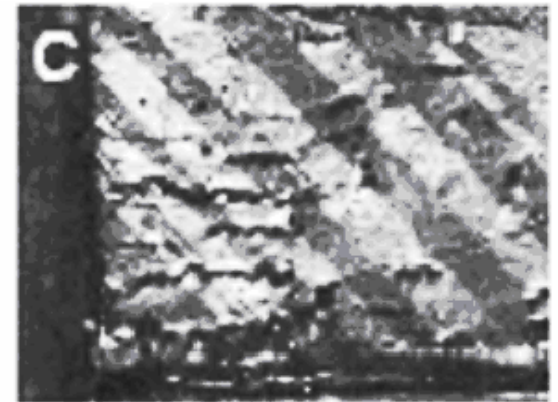
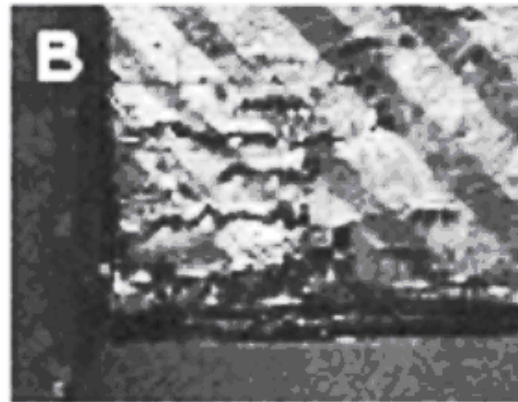
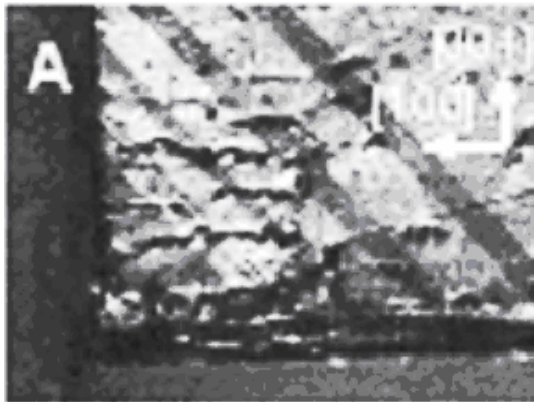
Part III: Experimental and Numerical Results





Laboratory experiment:

Martensite/martensite transformation at 20°C.
Change of microstructure under compression

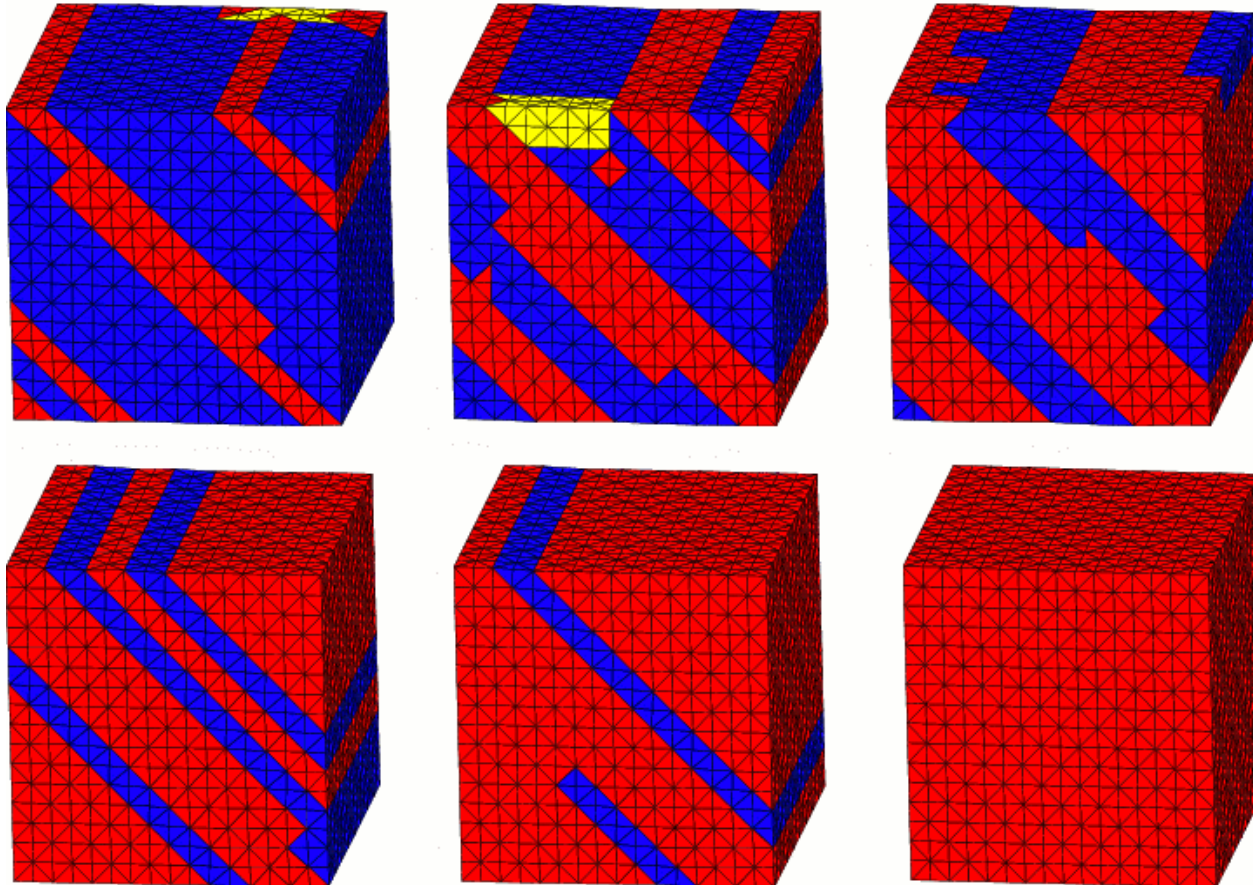




Numerical simulation:

Martensite/martensite transformation at 20°C.

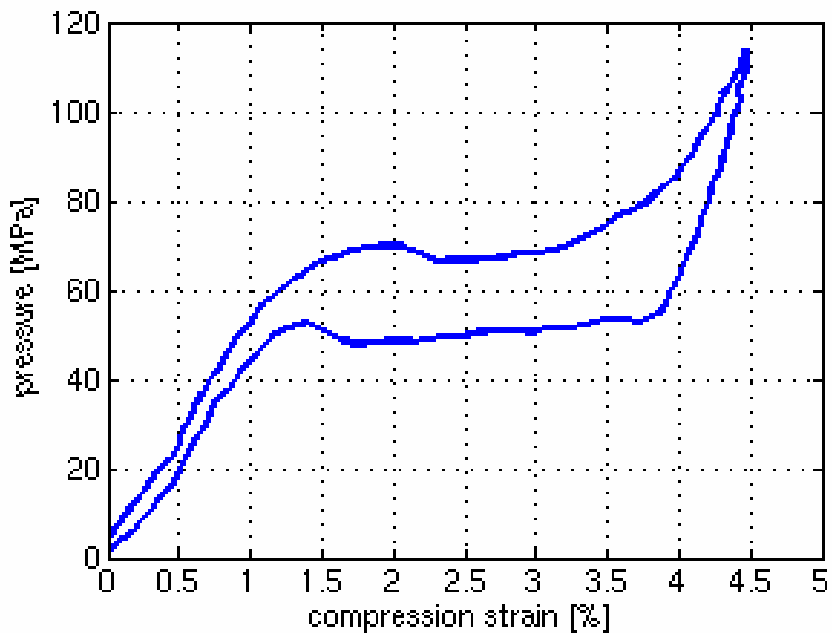
Change of microstructure under compression



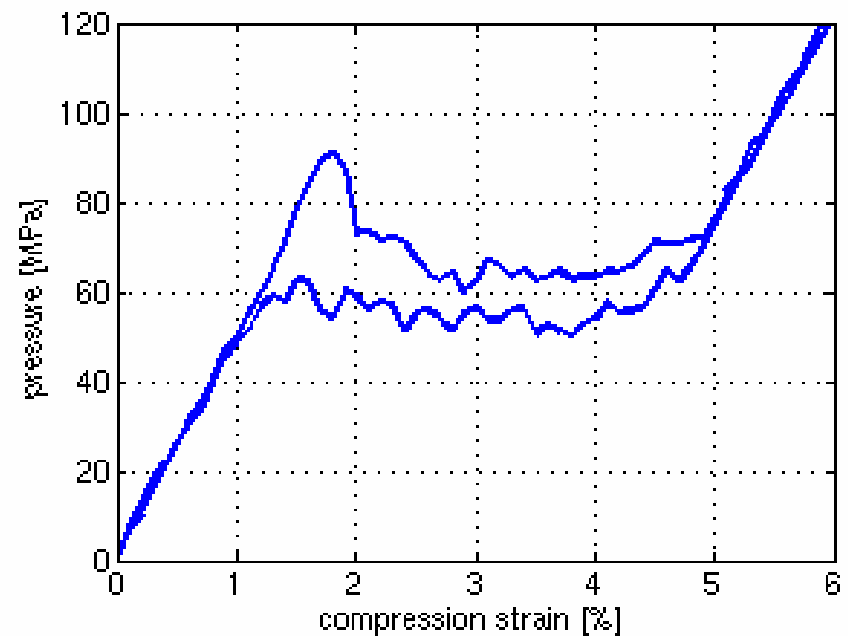


Stress-strain diagram for compression experiment:

Austenite/martensite transformation at 50°C



laboratory experiment

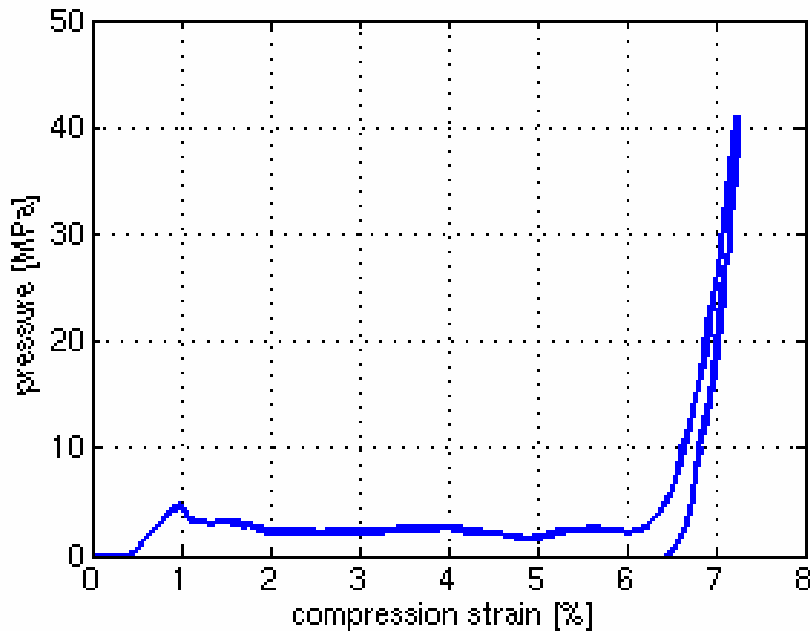


numerical simulation

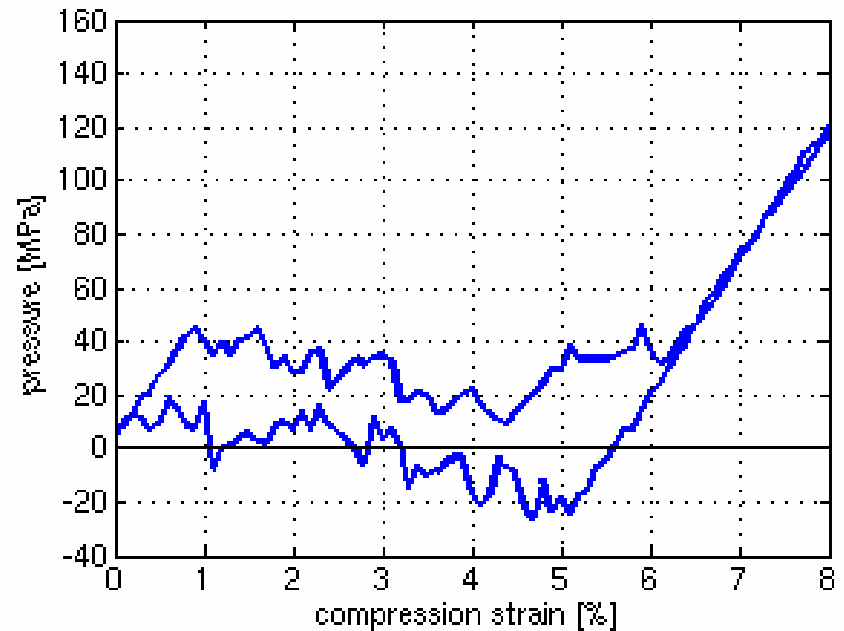


Stress-strain diagram for compression experiment:

Martensite/martensite transformation at 20°C



laboratory experiment



numerical simulation



Up to now: compression experiments.

Question: What happens under tension?

- **Laboratory experiment:**

Specimen needs to be fixed to loading machine.

But super-strong and rigid glue etc. not available.

Tension experiment in laboratory **impossible**.

- **Numerical simulation:**

Tension loading is no problem.

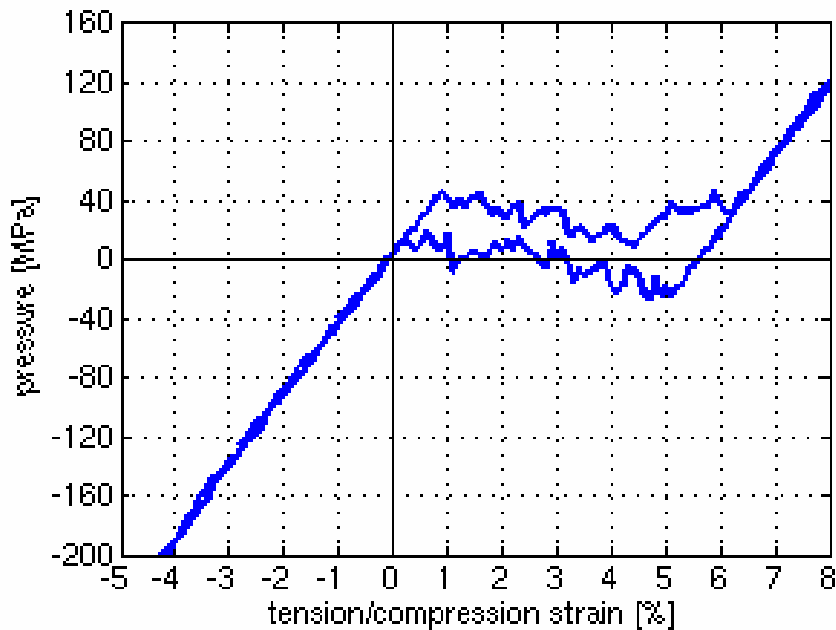
Model parameters have already been fitted for compression.

→ Use it for tension experiment as well

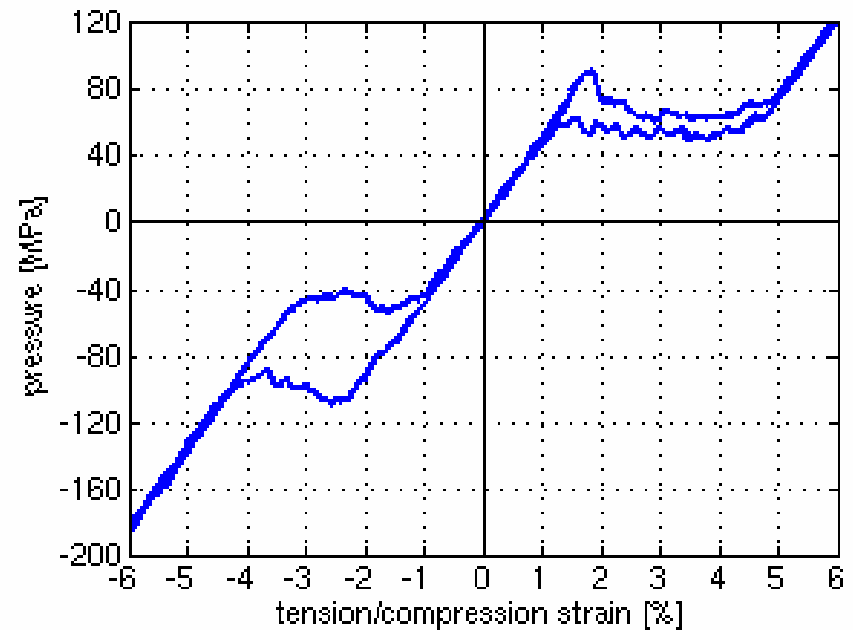
→ **Prediction of SMA behavior** under tension!



Numerical simulation of
compression and **tension** experiment:
Prediction of behavior of our NiMnGa specimen.



M/M transformation
at 20°C



A/M transformation
at 50°C